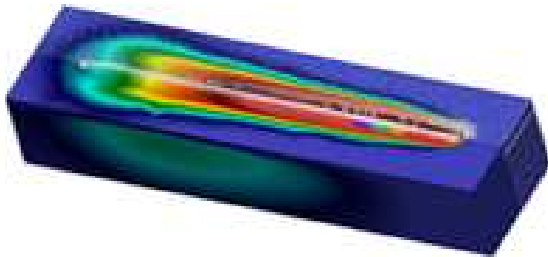
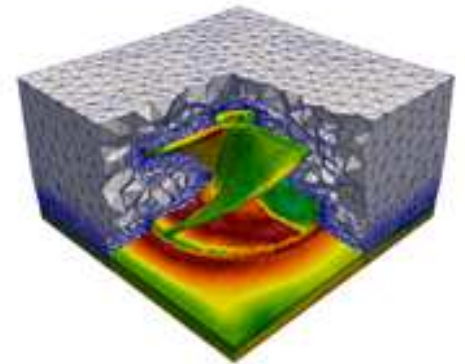


Thermo-mechanical modeling of additive manufacturing: continuum or particles with level set formulation applied to track and part scale simulations

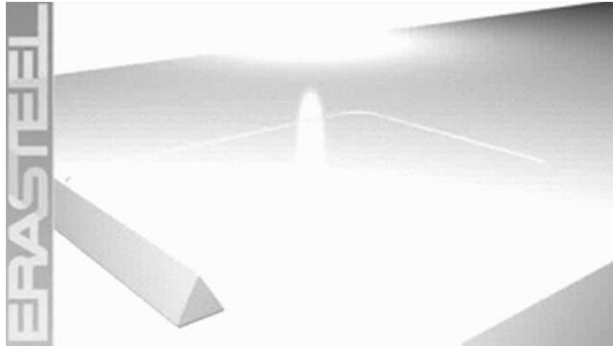


Yancheng Zhang,
Qiang Chen,
Gildas GUILLEMOT,
Michel Bellet,
Charles-André GANDIN

2MS, CEMEF, MINES ParisTech
E-mail: yancheng.zhang@mines-paristech.fr
Forum Teratec, 2019



Introduction: additive manufacturing selective laser melting (SLM)



Advantages:

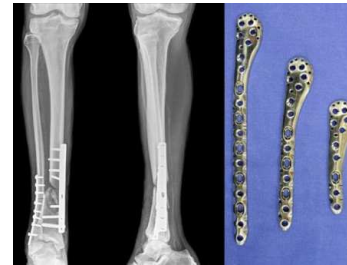
- Complex part geometry
- Variety of products and materials
- No time gap between design and prototyping
- Less waste than processes by subtraction

...



Jet engine
(site

[Scienceinpublic])



Implant
(Titanium, site
[Farinia])



Resistojet heat
exchanger
(316L, [Romei2017])

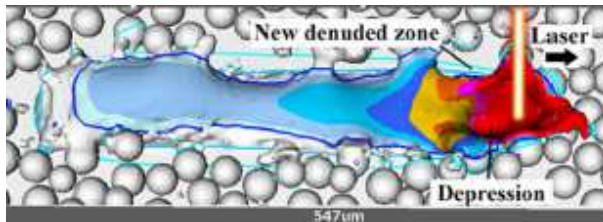
Challenges:

- Difficult to control
- Time consuming
- Defects (cracks, deformation, poor surface quality ...)

I. Introduction: 3 scales for SLM numerical modeling

3

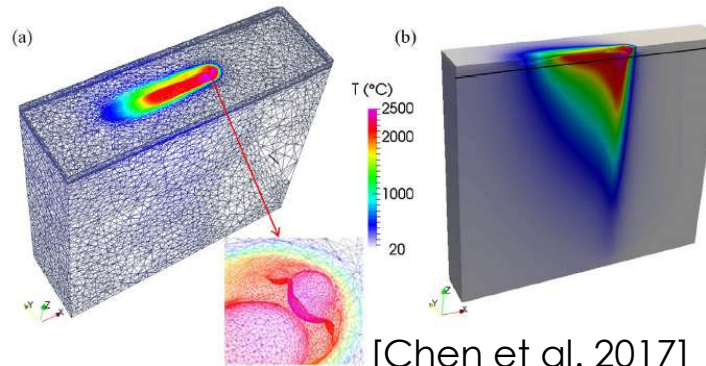
Powder particles « microscopic »



[Khairallah et al. 2016]

- **Interaction laser** – powder particles
- **Denudation phenomenon**
- **Formation of melted zone, porosities**
- **Influence of powder granulometry distribution**
- **Very high computing time**

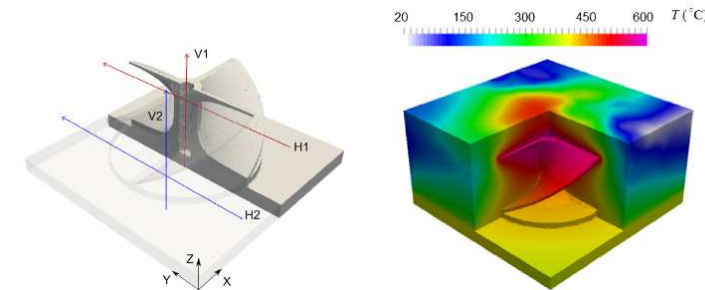
Track scale – Continuous powder bed « mesoscopic »



[Chen et al. 2017]

- **Interaction laser** – powder bed
- **Shape of elementary deposits** (tracks)
- Formation of **microstructure in extremely fast solidification and cooling**
- Formation of **stress** in wake of the melted zone
- Occurrence of defects: **balling, hot tearing**

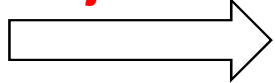
Part scale « macroscopic »



[Zhang et al. 2017]

- Energy and material inputs: simplified
- **Temperature** distribution
- Distributions of **distortions and stress**, during and after processing
- Thermal and mechanical role of **support** structures
- Occurrence of defects: **cold** cracking

Objective



Numerical strategy developed in CEMEF to simulate the SLM process with **powder bed** at the **meso-** and **macro-scale** by **level set** and **finite element** methods

Khairallah, Anderson, Rubenchik, King, Laser powder-bed fusion additive manufacturing: physics of complex melt flow and formation mechanisms of pores, spatter, and denudation zones, *Acta Materialia* (2016)

Chen, Guillemot, Gandin, Bellet, Three-dimensional finite element thermomechanical modeling of additive manufacturing by selective laser melting for ceramic materials, *Additive Manufacturing* (2017)

Zhang, Guillemot, Bernacki, Bellet, Macroscopic thermal finite element modeling of additive metal manufacturing by selective laser melting process, *Comp. Meth. Appl. Mech. Eng.* (2018)

Presentation plan

- ▶ **Introduction**
- ▶ **Thermal and mechanical solvers in developed meso- and macro-scale models**
- ▶ **Meso-scale model at the track level**
 - ❑ Interaction of laser-material
 - ❑ Hydrodynamics
 - ❑ Balling effect
 - ❑ Thermomechanical analysis
- ▶ **Macro-scale model at part level**
 - ❑ Strategy of fraction of layer
 - ❑ Thermomechanical analysis
 - ❑ Inherent strain rate method
- ▶ **Conclusions**
- ▶ **Perspectives**

Thermal solver: meso-scale model

5

► Heat transfer:

$$\frac{\partial\{\rho h\}}{\partial t} + \nabla \cdot (\{\rho h\}\mathbf{u}) - \nabla \cdot (\{\lambda\}\nabla T) = \dot{q}_L - \dot{q}_r$$

Heat loss

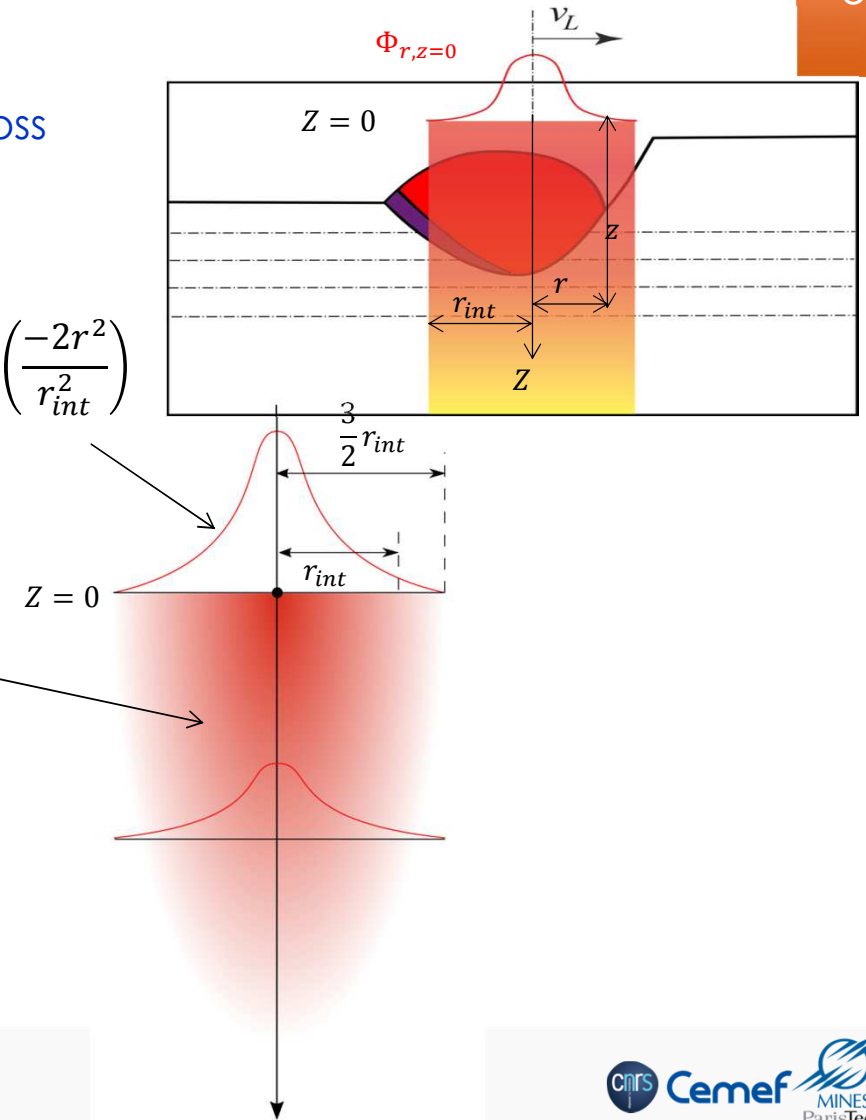
► Laser heat source : \dot{q}_L

- Gaussian surface distribution $\phi_{r,z=0}$
- Volume heat source based on Beer-Lambert law

$$\frac{d\phi}{dz} = -\alpha\phi \quad \Rightarrow \quad \phi(r,z) = \phi_{r,z=0} \exp\left(-\int_0^z \alpha dz\right)$$

$$\dot{q}_L = -\frac{d\phi}{dz} = \phi_{r,z=0} \cdot \alpha \cdot \exp\left(-\int_0^z \alpha dz\right)$$

$$\phi_{r,z=0} = (1 - R) \frac{2P_L}{\pi r_{int}^2} \exp\left(\frac{-2r^2}{r_{int}^2}\right)$$



Meso-scale modeling – track scale: approach

Multidomain multiphase level set representation [Hagedorn et al., 2010 and Chen et al., 2017]

► **Multidomain system: D_i , $i = \{1, 2\}$**

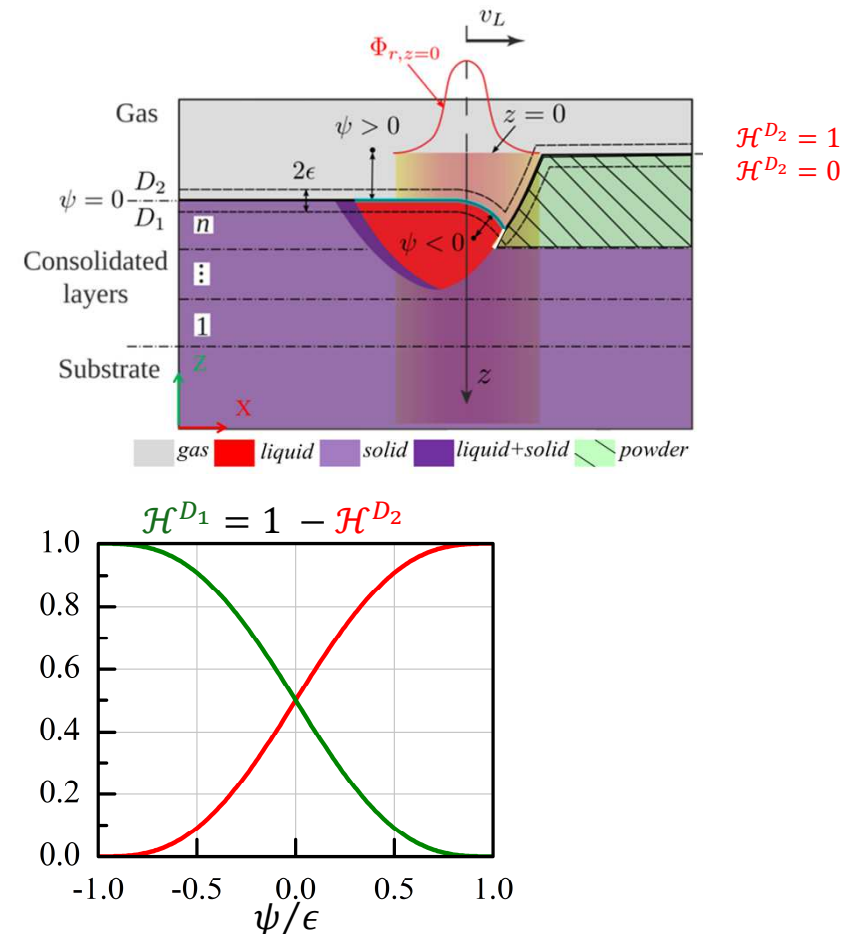
- ❑ Single phase domain D_2 : phase *gas*
- ❑ Multiphase domain D_1 : phase α_l
- ❑ Powder is considered as continuous

► **Interdomain boundary D_1/D_2**

- ❑ Level set function ψ , i.e. signed distance from boundary D_1/D_2 with $\psi = 0$
- ❑ Heaviside function with half-interface thickness ϵ

► **Volume averaging**

- ❑ Within the system: $\chi = \sum_{D_i} \mathcal{H}^{D_i} \langle \chi \rangle^{D_i}$
- ❑ Within each domain D_i : $\langle \chi \rangle^{D_i} = \sum_{\alpha_l} g_{D_i}^{\alpha_l} \chi_{D_i}^{\alpha_l}$



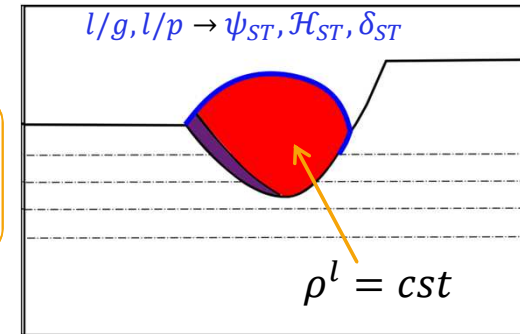
Meso-scale modeling: hydrodynamics

► Momentum equation:

$$\rho \left(\frac{\partial \mathbf{u}}{\partial t} + \mathbf{u} \cdot \nabla \mathbf{u} \right) - \nabla \cdot \underline{\underline{\sigma}} = \rho \mathbf{g} + \mathbf{f}$$

$$\nabla \cdot \mathbf{u} = \dot{\theta}$$

Buoyancy force &
recoil pressure
not studied



► Compressible Newtonian behavior

$$\underline{\underline{\sigma}} = 2\mu \left(\underline{\underline{\dot{\epsilon}}} - \frac{1}{3} \text{tr}(\underline{\underline{\dot{\epsilon}}}) \underline{\underline{I}} \right) - p \underline{\underline{I}} \quad \text{and} \quad \text{tr}(\underline{\underline{\dot{\epsilon}}}) = \nabla \cdot \mathbf{u} = \dot{\theta}$$

► Surface tension \mathbf{f}_s [Hysing, 2005, Khalloufi et al., 2016] and Marangoni force \mathbf{f}_m

$$\mathbf{f} = \mathbf{f}_s + \mathbf{f}_m$$

$$\mathbf{f}_s = \delta_{ST} \gamma \kappa \mathbf{n}$$

$$\mathbf{f}_m = \delta_{ST} \frac{\partial \gamma}{\partial T} \nabla_s T$$

CSF method [Brackbill et al., 1992] is applied with $\delta_{ST} = \partial \mathcal{H}_{ST} / \partial \psi_{ST}$

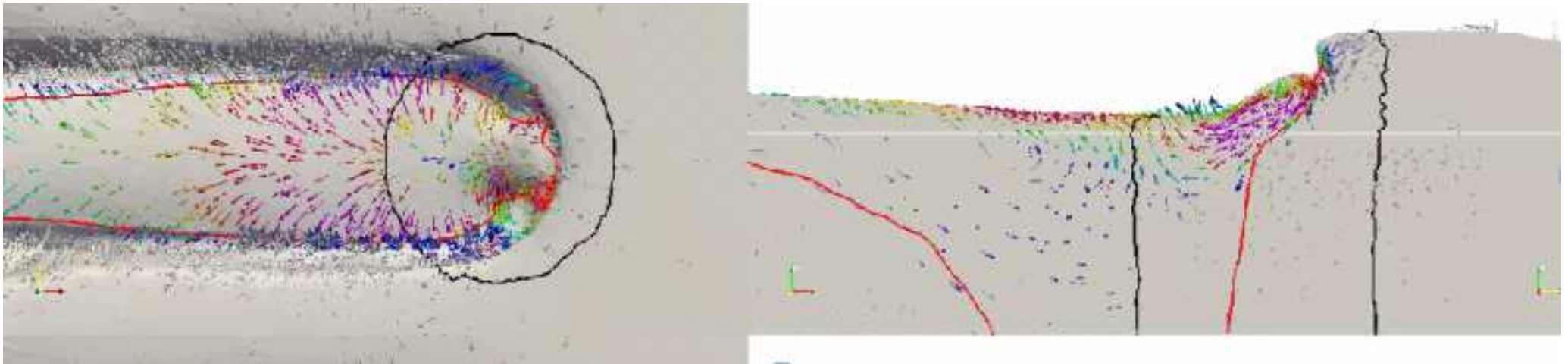
Hysing, International Journal for Numerical Methods in Fluids, 51:659-672, 2005.

Khalloufi, Mesri, Valette, Massoni, Hachem, Computer Methods in Applied Mechanics and Engineering, 307: 44-67, 2016.

Brackbill, Kothe, Zemach, Journal of Computational Physics 100 (1992) 335-354

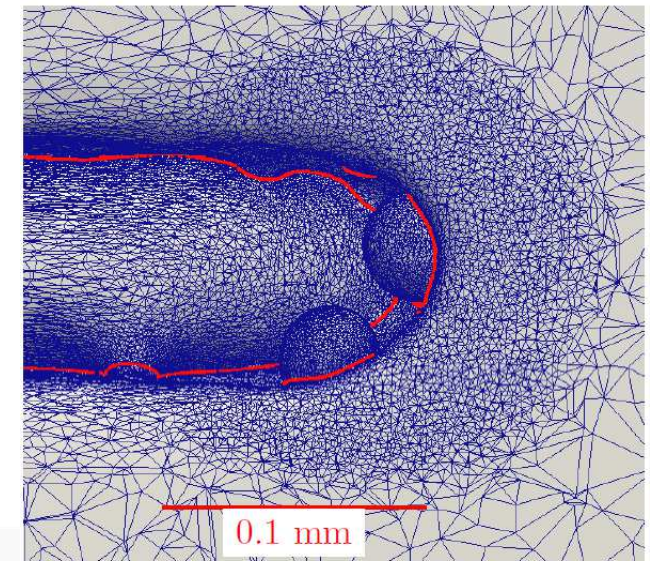
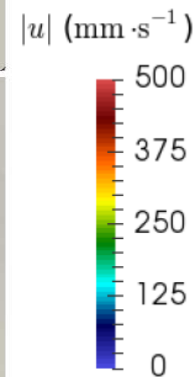
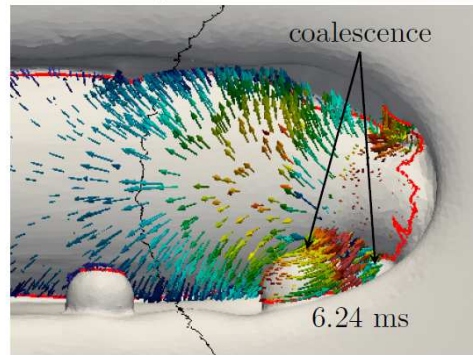
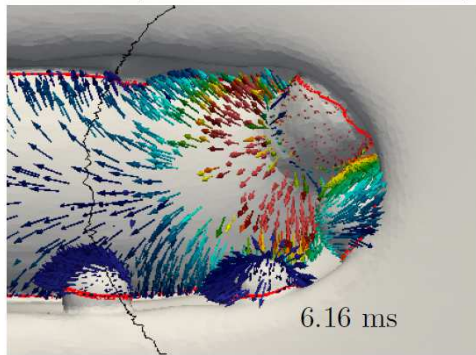
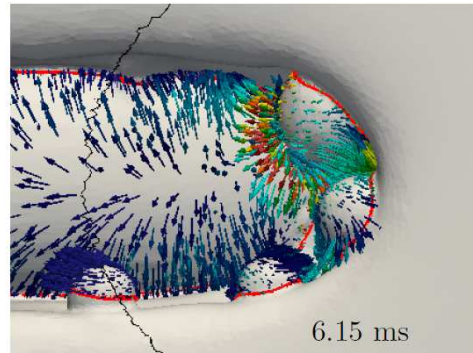
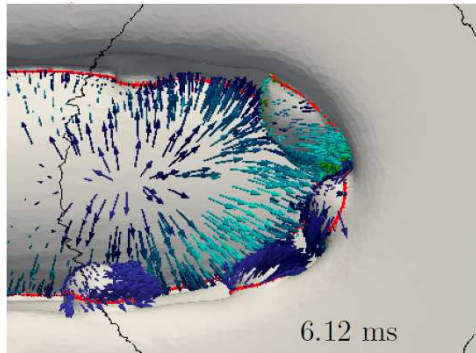
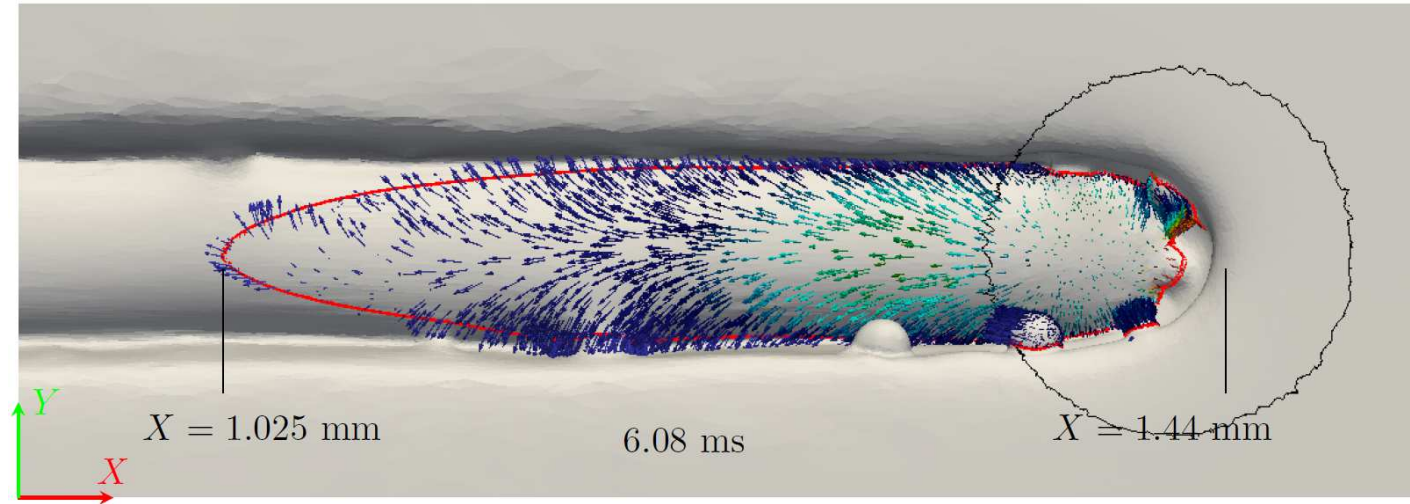
Meso-scale: fluid flow

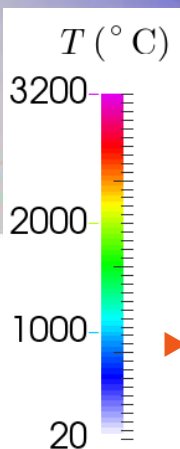
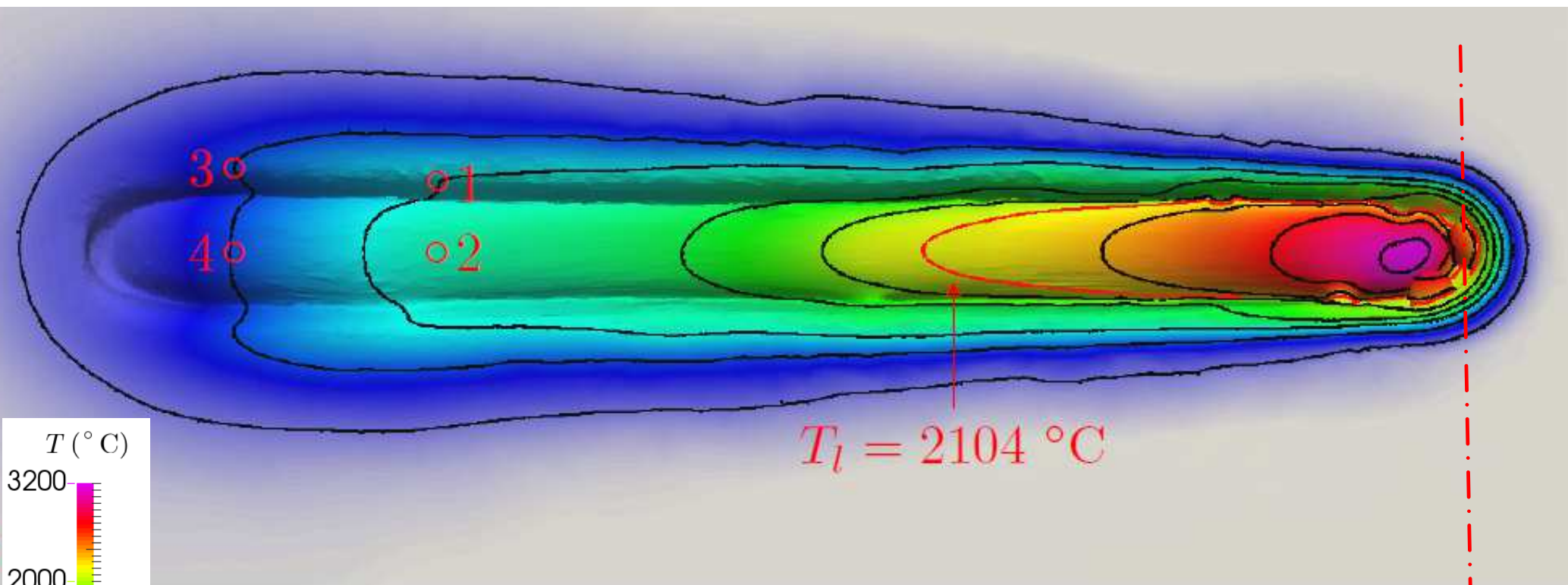
- ▶ **Heat transfer solver:** fusion, solidification, non-linear (latent heat)...
- ▶ **Navier-Stokes solver:** stabilized, surface tension, Marangoni...
- ▶ **Coupling of both:** convection effects



Fluid flow

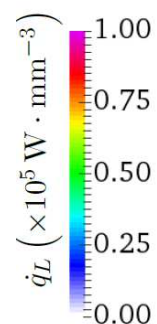
- ▶ **Heat transfer solver:**
fusion, solidification, non-linear (latent heat)...
- ▶ **Navier-Stokes solver:**
stabilized, surface tension, Marangoni...
- ▶ **Coupling of both:**
convection effects



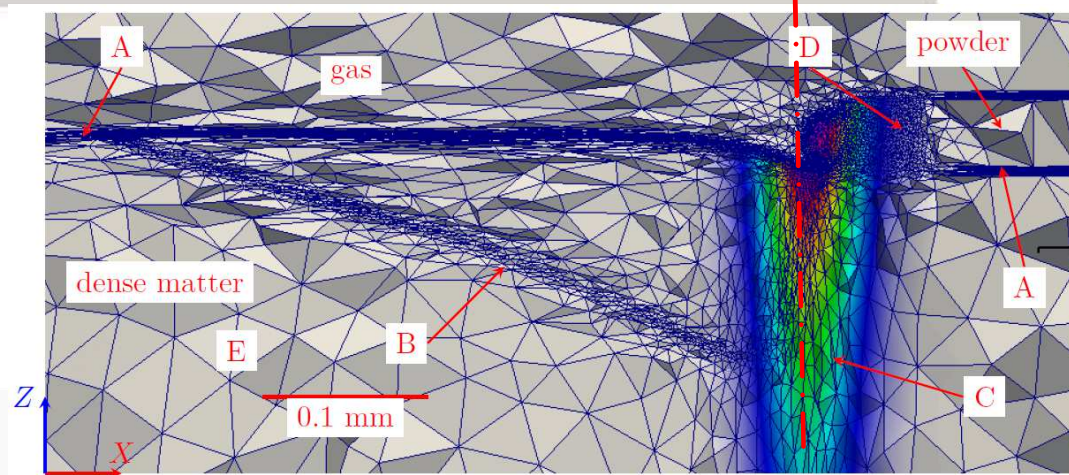


► **Mesh adaptation governed by:**

- Density (\rightarrow zone A)
- Liquid fraction (\rightarrow zone B)
- Heat source (\rightarrow zone C)
- Background mesh size : $50\text{ }\mu\text{m}$



Thermal analysis and mesh

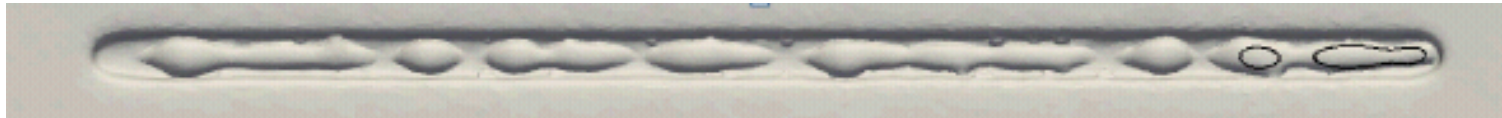


Meso-scale: balling effect

- Destabilization of the liquid bead by increasing the speed of the laser

$$P_L = 84 \text{ W}$$

V550



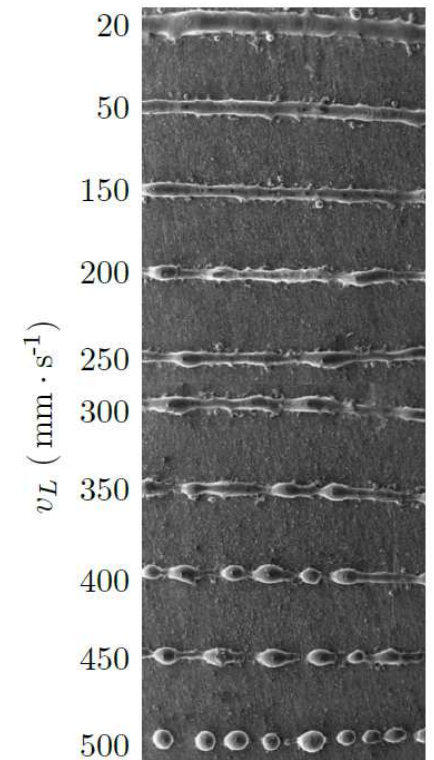
Black contour=
temperature at the end
of solidification
($T = 2004 \text{ }^{\circ}\text{C}$)



V600



V800



Balling effect (316L stainless steel)

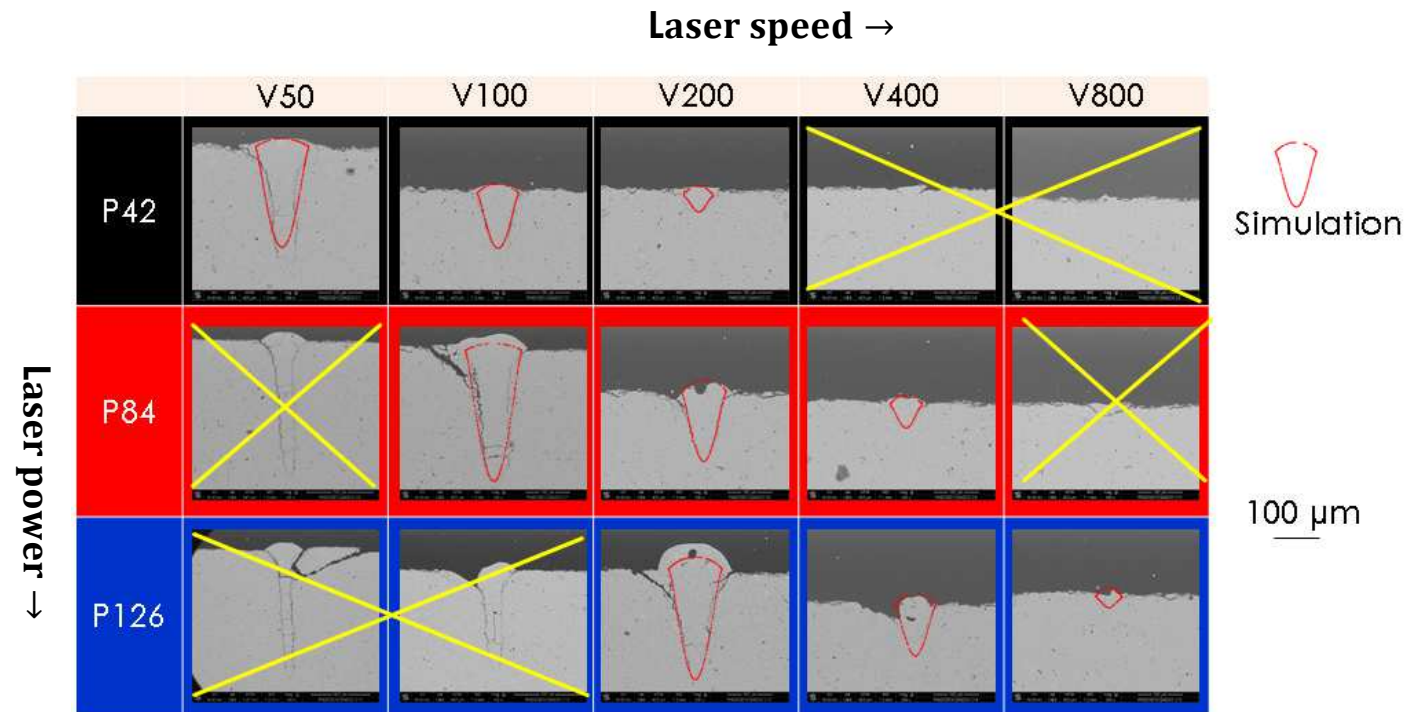
Li et al., Balling behavior of stainless steel and nickel powder during SLM, Int J Adv Manuf Tech 59 (2012)

Meso-scale: track shape

12

- ▶ Simulation vs Experimental
- ▶ PhD of Liliana Moniz, Centre des Matériaux, MINES ParisTech, Evry

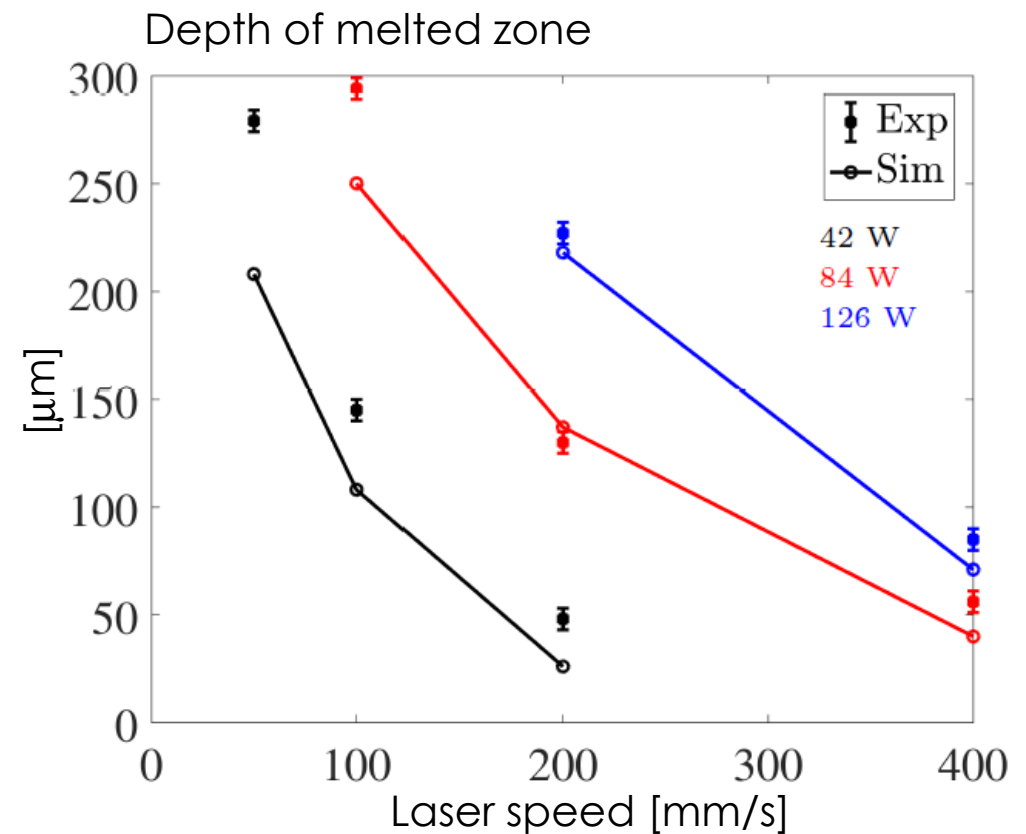
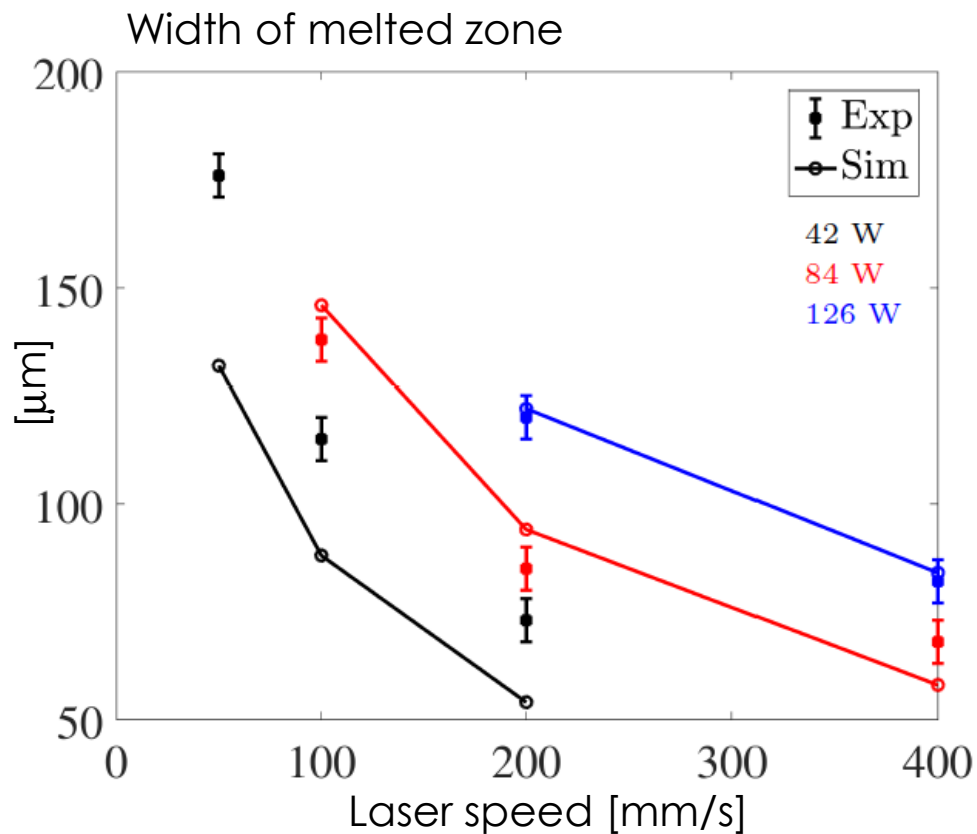
- ❑ Alumina powder $D_V(50) = 14.9 \mu\text{m}$ doped with 1% carbon (in mass)
- ❑ Undoped quasi-dense alumina substrate
- ❑ Layer thickness: $50 \mu\text{m}$
- ❑ Porosity of powder bed: 60%
- ❑ Laser Yb:YAG, $\lambda_L = 1070 \text{ nm}$, $P_L = 168 \text{ W max}$



Validation - form of melted zone: simulation vs experiment

13

► Calibration of 3 parameters: R_{int} , α_s et α_l and applied simulations



Mechanical solver: meso- and macro-scale models

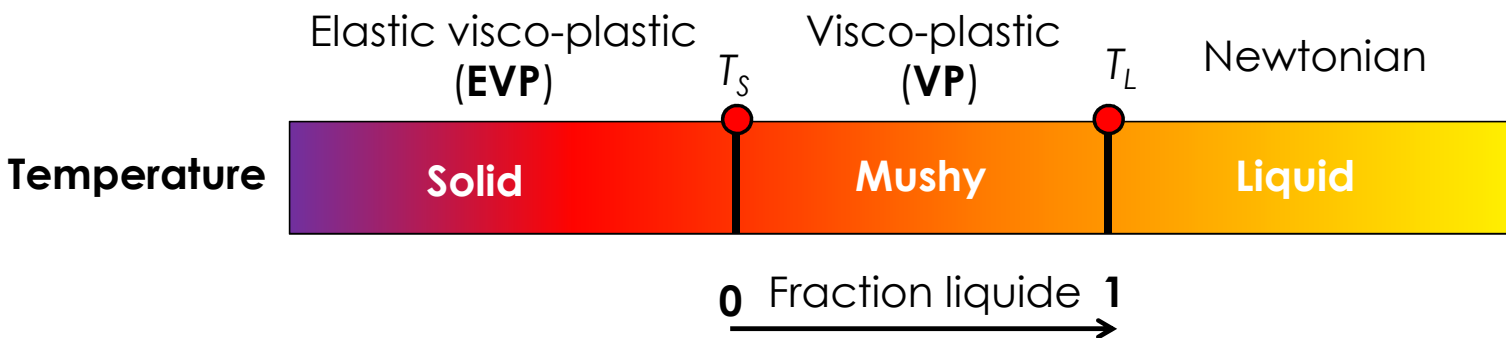
14

► Equilibrium equation:

$$\nabla \cdot \underline{\underline{\sigma}} + \rho \underline{\underline{g}} = \underline{\underline{0}}$$

► State and constitutive model:

▣ Constructed part:



▪ EVP

$$\underline{\underline{\dot{\epsilon}}} = \underline{\underline{\dot{\epsilon}}}^{el} + \underline{\underline{\dot{\epsilon}}}^{vp} + \underline{\underline{\dot{\epsilon}}}^{th}$$

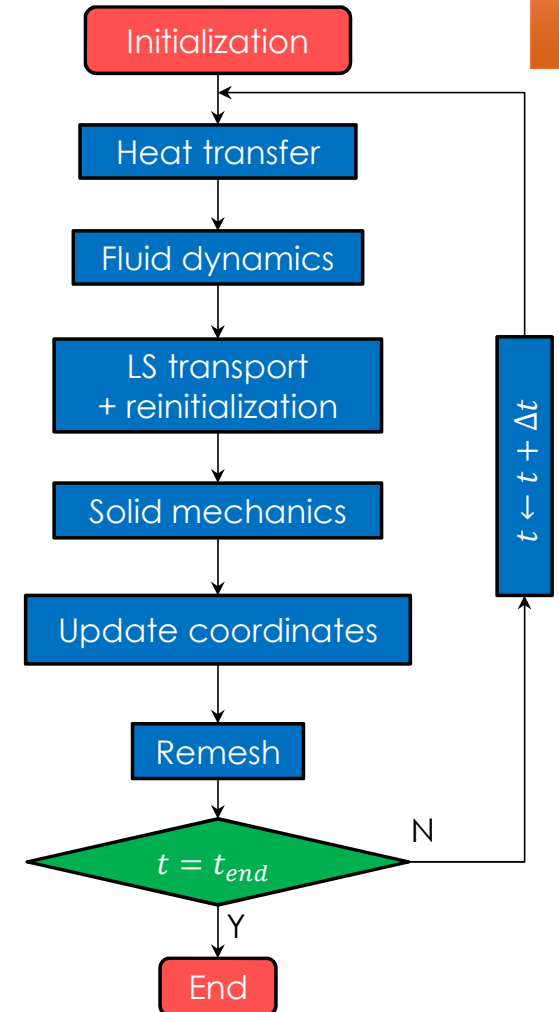
$$\bar{\sigma} = \sigma_Y + K(\sqrt{3})^{m+1} \bar{\epsilon}^n \dot{\epsilon}^m$$

▪ VP

$$\underline{\underline{\dot{\epsilon}}} = \underline{\underline{\dot{\epsilon}}}^{vp} + \underline{\underline{\dot{\epsilon}}}^{th}$$

$$\bar{\sigma} = K(\sqrt{3})^{m+1} \dot{\epsilon}^m$$

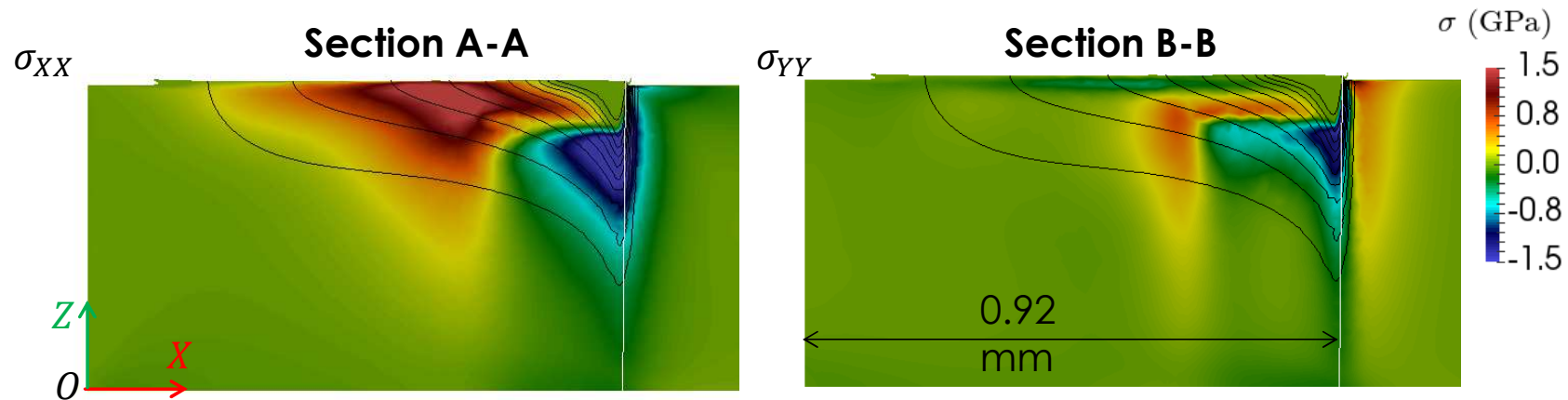
▣ Non-melted powder and gas: Newtonian



Meso-scale: hot stress during the process

15

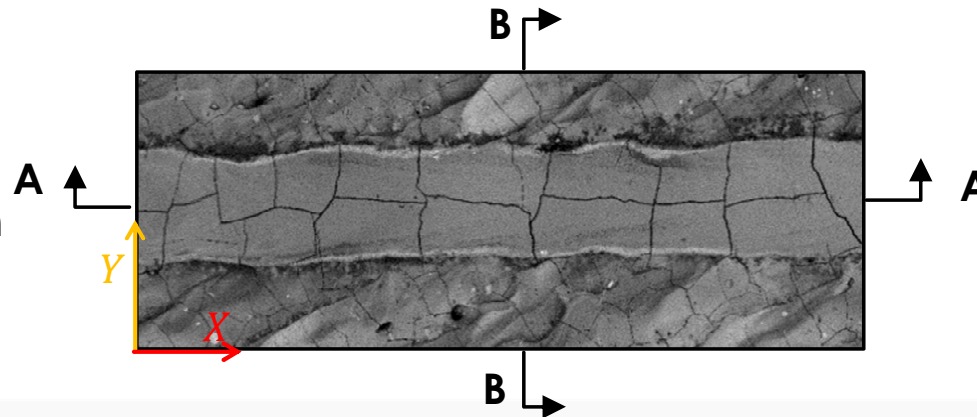
$$P_L = 84 \text{ W} \quad v_L = 200 \text{ mm} \cdot \text{s}^{-1}$$



$t = 3.6 \text{ ms}$

Iso-contours: $T = 300 \sim 2100 \text{ }^\circ\text{C}$ with $\Delta T = 200 \text{ }^\circ\text{C}$

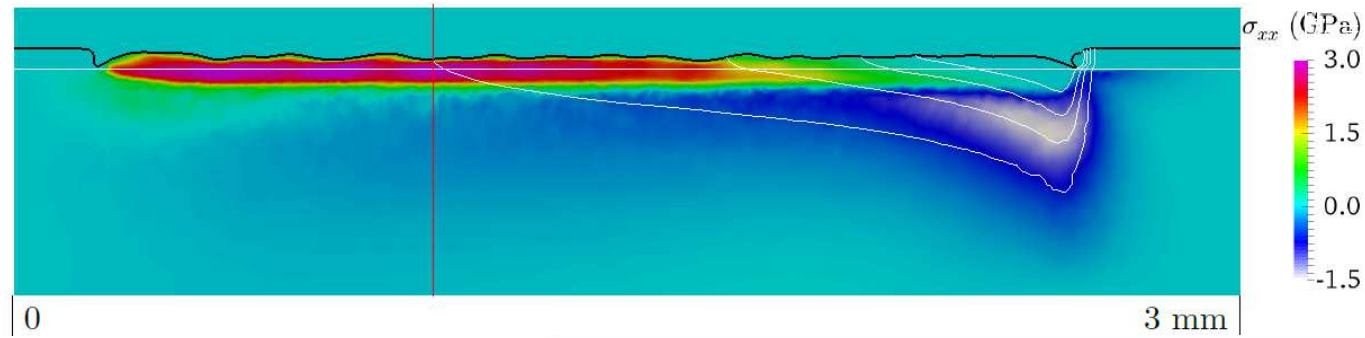
Experimental observation
(top view)



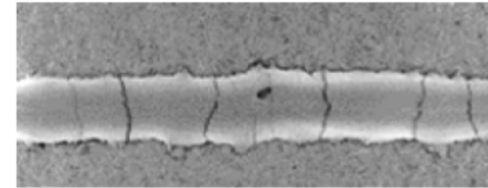
Effect of auxiliary heating

16

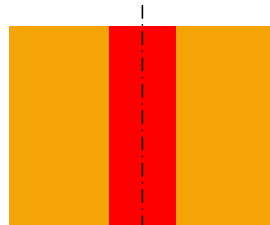
Single laser



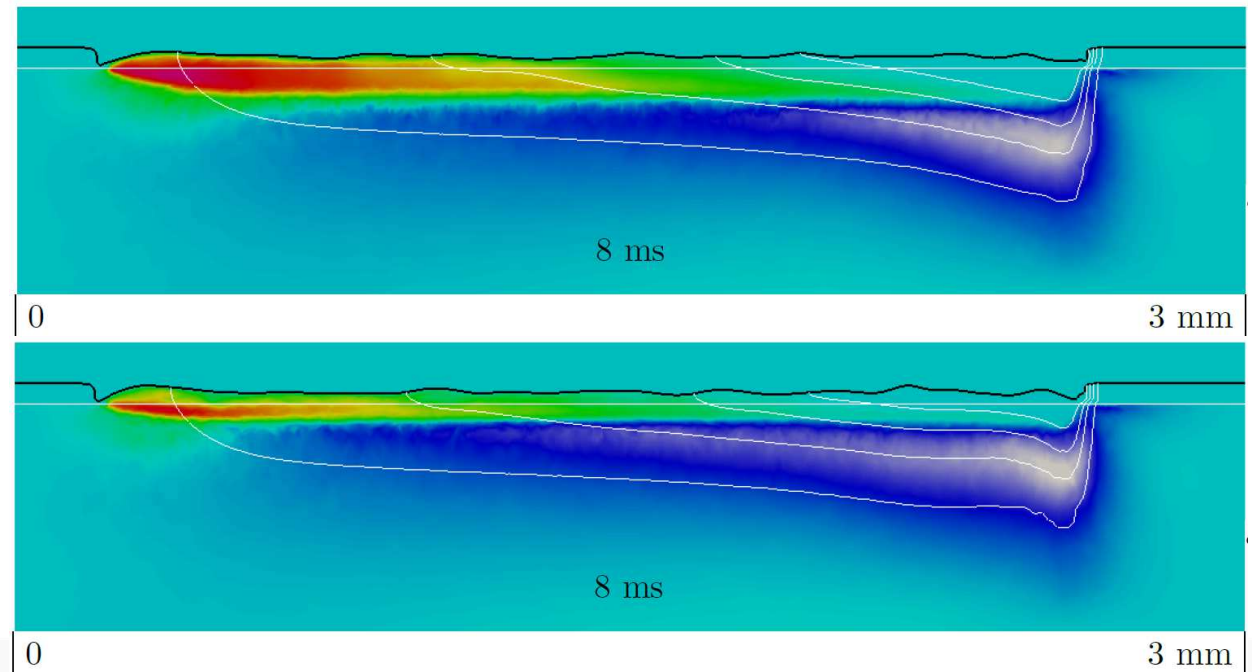
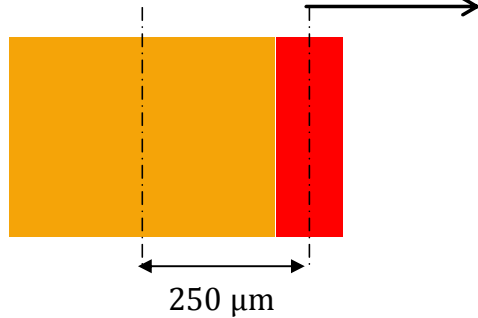
400 μm



2 lasers, coaxial



2 lasers, tandem



Axial stress σ_{xx}

Micro-scale model: explicit particles of powder

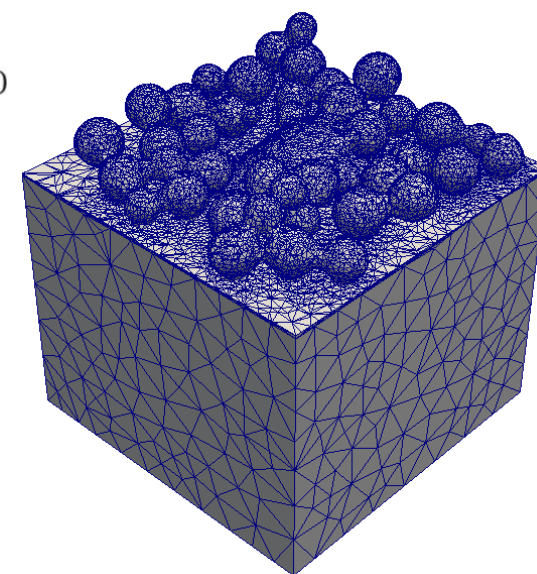
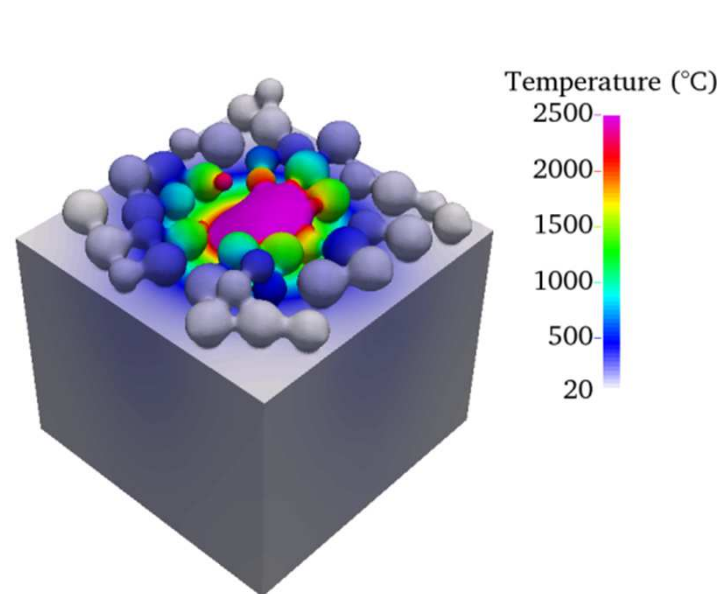
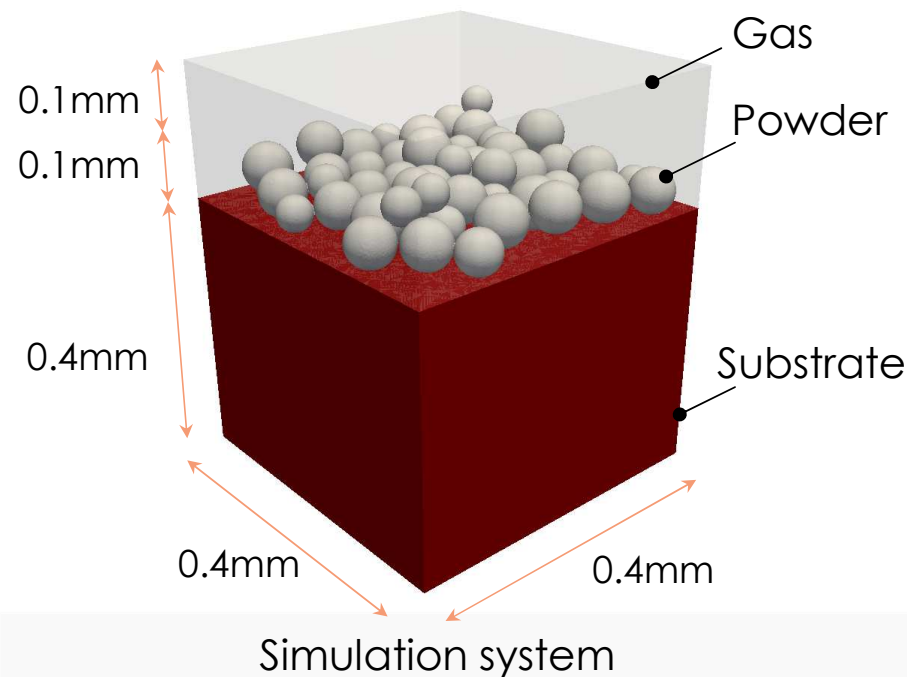
► **Master thesis: Yuwei SHAN (co-supervision), SPEIT, Shanghai Jiaotong University.**

Supervisors: CEMEF: Yancheng ZHANG and Michel BELLET

SPEIT: Jin YU and lingxuan SHAO

The aim is to study the melting and coalesce of the particles by both experimental and simulation aspects:

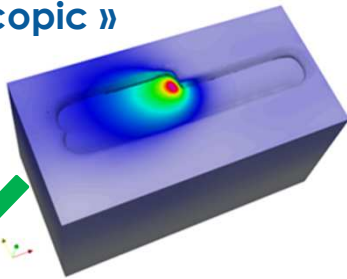
- Predict the porosity of the constructed part due to inefficient laser power
- Validate the selected the model of heat source (**Beer-Lambert law** at the first step).



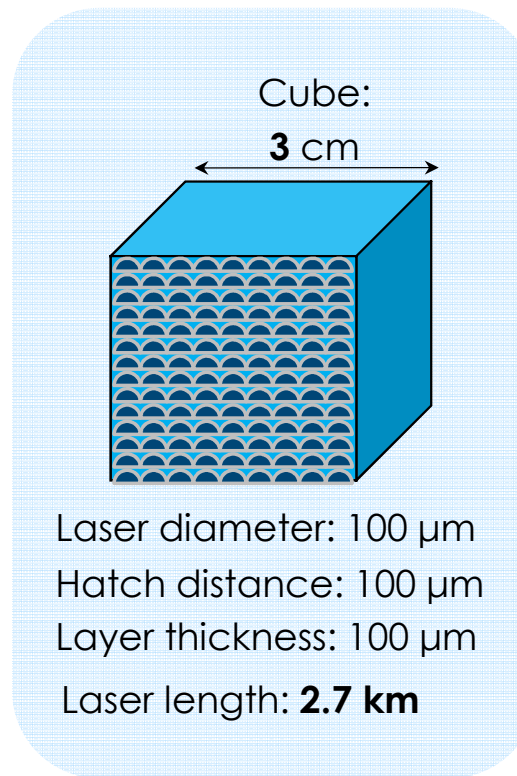
Modelling strategy for scale selection

► Calculation in 28 cores with cluster

Track scale – Continuous powder bed
« mesoscopic »



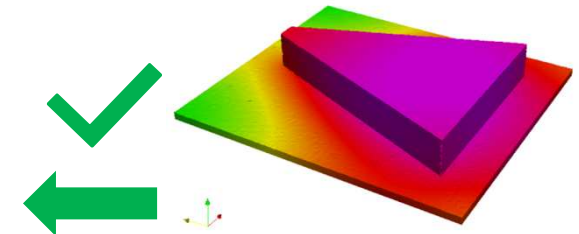
Comp. time:
1300 day/m
(~**30** h / mm)



Cube:
3 cm

Laser diameter: 100 μm
Hatch distance: 100 μm
Layer thickness: 100 μm
Laser length: **2.7 km**

Part scale
« macroscopic »



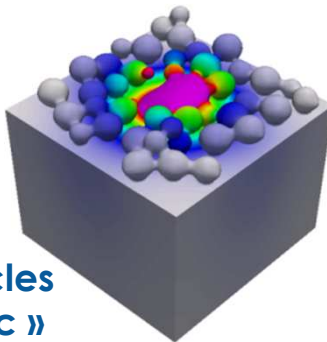
Comp. time:

Fraction of layer: **0.05** day/m
Layer by layer: $0.05/n_f$ day/m
Super layer: $0.05/n_f/n_l$ day/m

n_f : number of fraction

n_l : number of layer

Powder particles
« microscopic »



Comp. time:
15000 day/m
(~**15** day / mm)

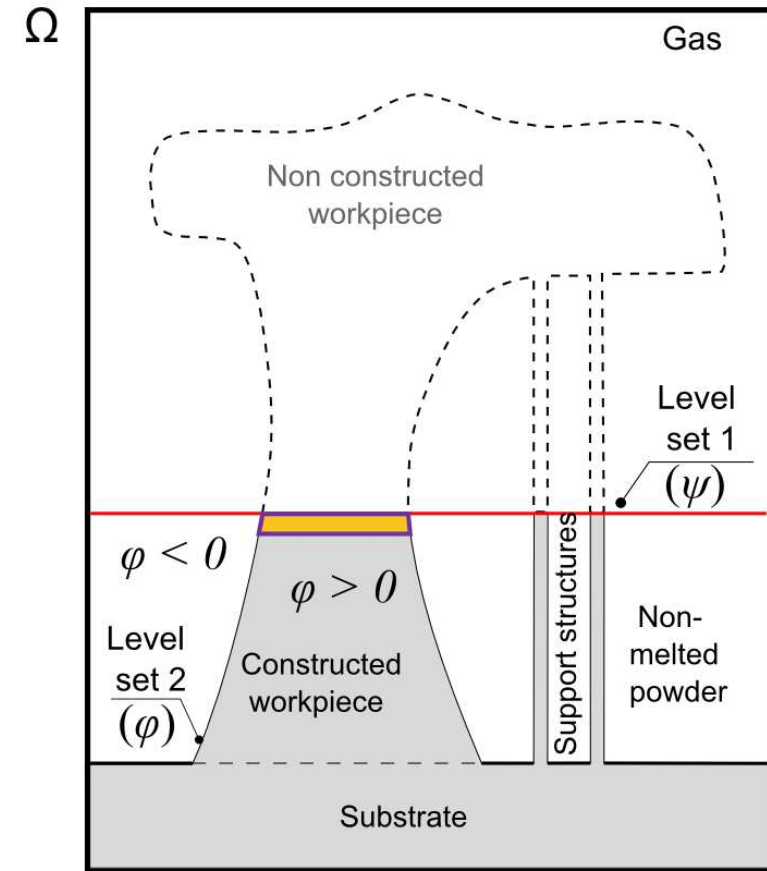
Mesoscopic : [Che18] Q. Chen, G. Guillemot, Ch.A. Gandin, M. Bellet, Numerical modelling of the impact of energy distribution and Marangoni surface tension on track shape in SLM of ceramic material, Add. Manuf. 21 (2018), 713-723 **Macroscopic** : [Zha18] Y. Zhang, G. Guillemot, M. Bernacki, M. Bellet, Macroscopic thermal finite element modeling of AM manufacturing by SLM process, Comp. Met. App. Mech. Eng. 331 (2018) 514-535

Macro-scale modeling: in part level

► Works of Yancheng Zhang

► Principles of the construction

- ❑ Starting point: **CAD** of the **part** to be built, completed by **the substrate** and the possible **supports**.
- ❑ Mesh of this **CAD**.
- ❑ Take a **background mesh** Ω and immerse the previous **CAD** mesh into it.
- ❑ Construct the conforme mesh at interfaces:
 - material / gas
 - constructed part/ non-exposed powder
- ❑ Over time, the **level set** ψ is updated progressively through the mesh to simulate the material deposition **layer by layer** or **fraction of layer**
- ❑ Resolve the **thermal** and **mechanical** problems in each time steps:
 - in the part under construction...
 - ... but also in the non-exposed powder



Schematic of Level Set

Thermal solver: macro-scale model

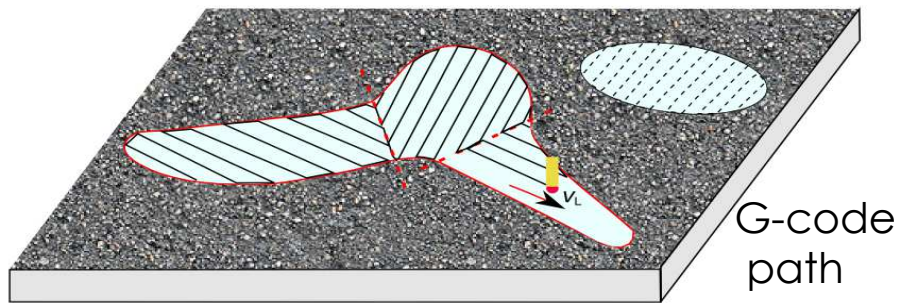
20

► Heat transfer:

$$\frac{\partial\{\rho C\}}{\partial t} - \nabla \cdot (\lambda \nabla T) = \dot{q}_L$$

ρC and λ are element-wise mixed properties

► Heat flux approximation for the fractions: \dot{q}_L

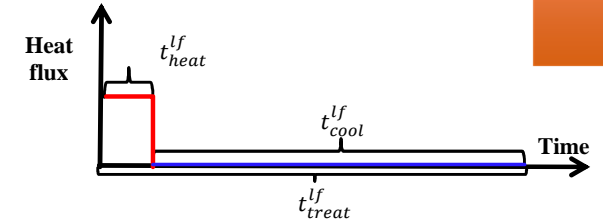


- Energy balance: surface heat source

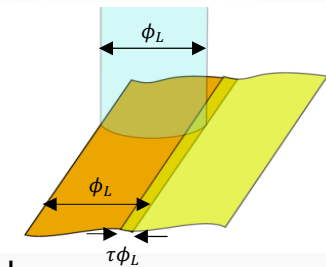
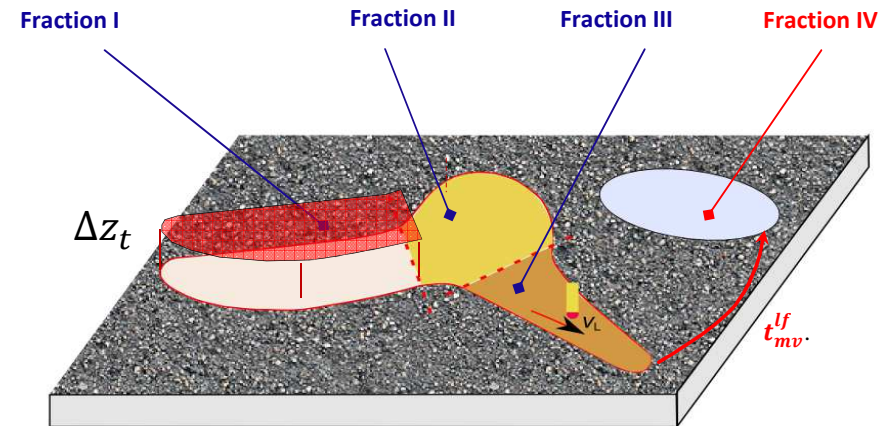
$$(1 - R)P_L t_{scan}^{lf} = S^{lf} \dot{q}_L^{s} t_{heat}^{lf}$$

- Homogeneous volume heat source

$$\dot{q}_L = \frac{\dot{q}_L^s}{\Delta z_t} = \frac{P_L(1 - R)}{S^{lf} \Delta z_t} \frac{t_{scan}^{lf}}{t_{heat}^{lf}}$$



Decomposition of the time interval

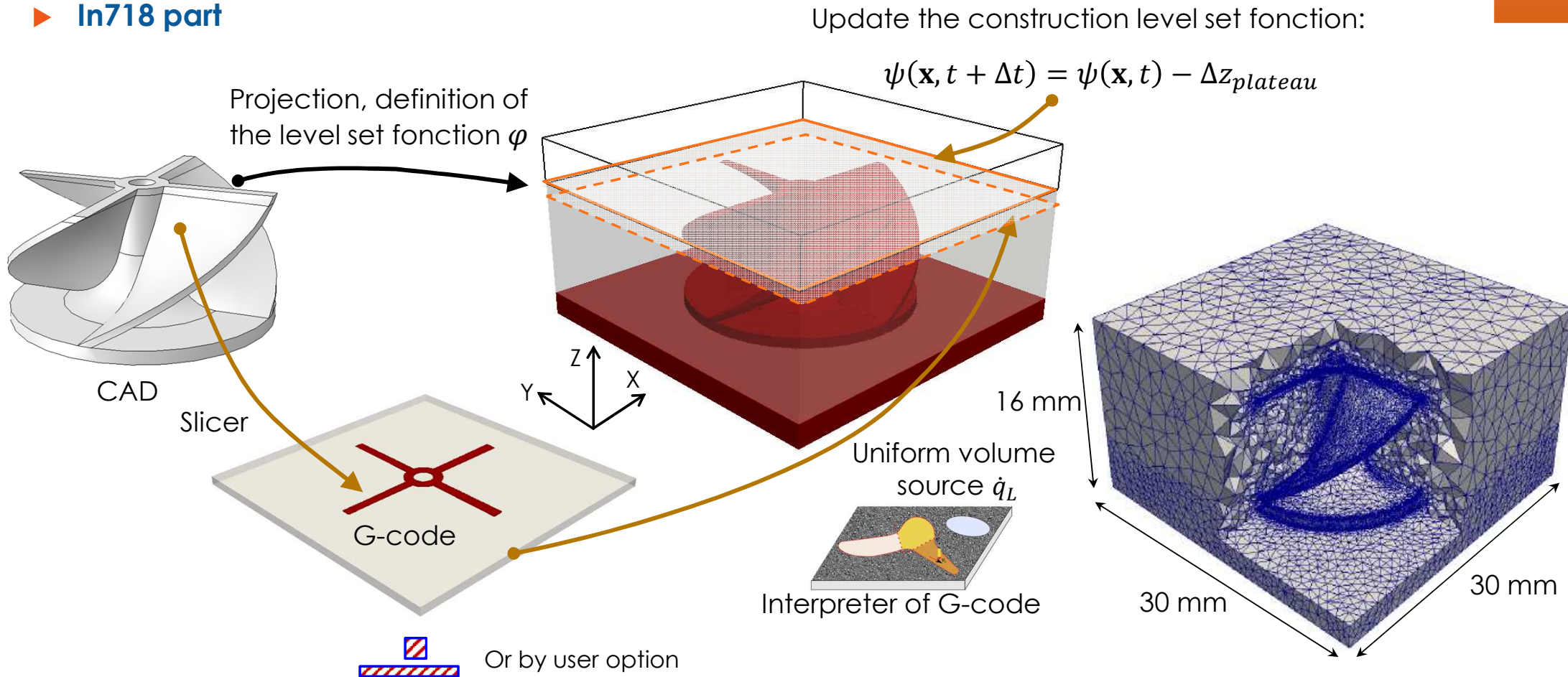


Laser scan intersection

Macro-scale modeling: application sample

21

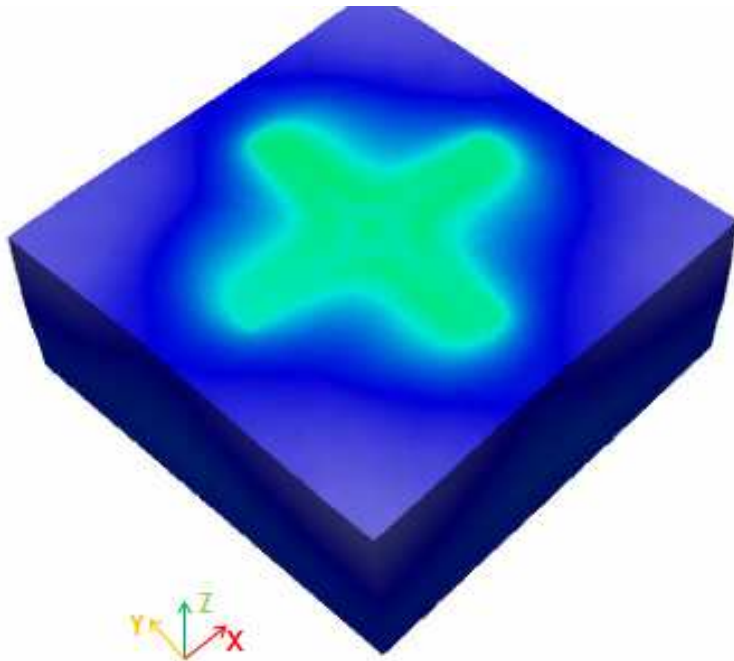
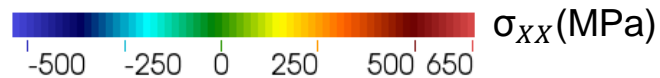
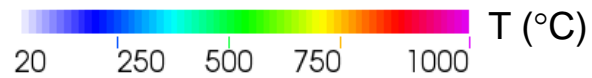
► In718 part



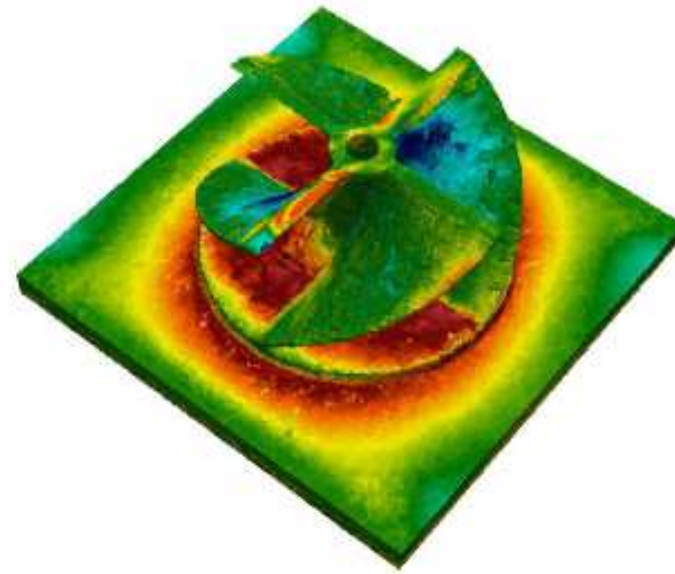
Macro-scale modeling: simulation results

22

► In718 part



Temperature: with non-exposed powder



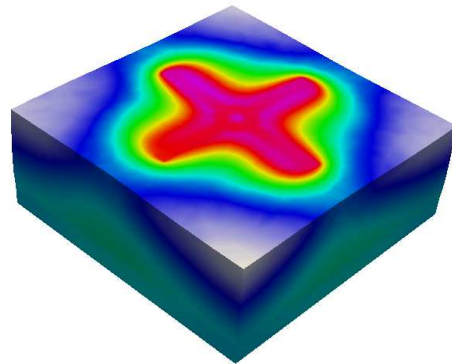
Stress: without non-exposed powder

Temperature and stress distribution during the construction process (50 layers)

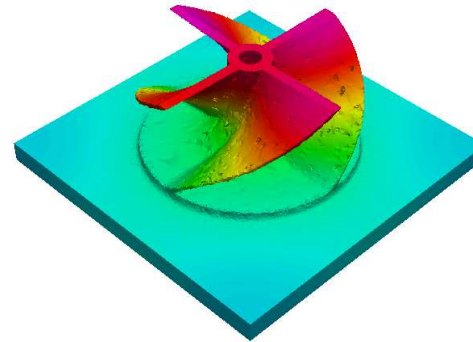
Macro-scale modeling: simulation results

23

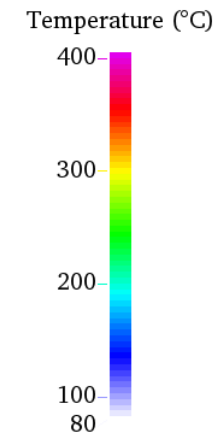
► Final construction: temperature



Part + non-exposed powder



Part



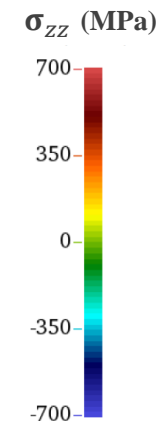
► Final construction + 2h: residual stress



Cool down to room temperature



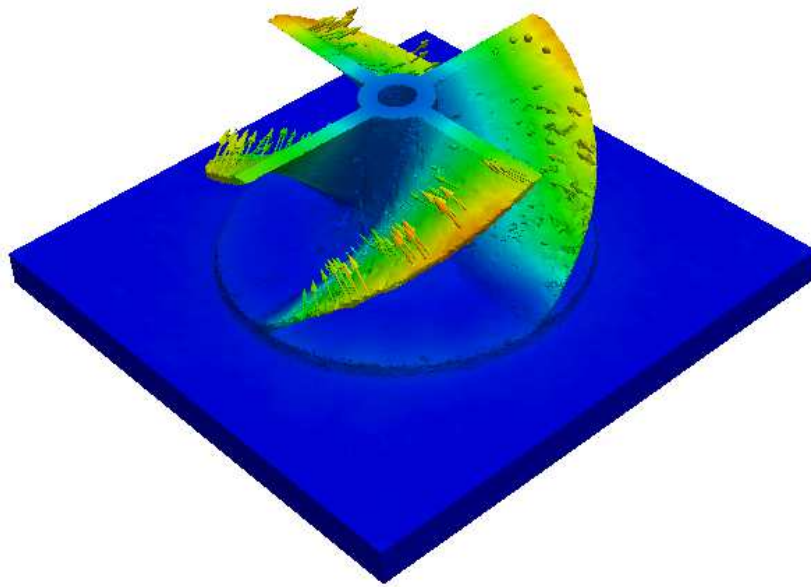
After removing from the substrate



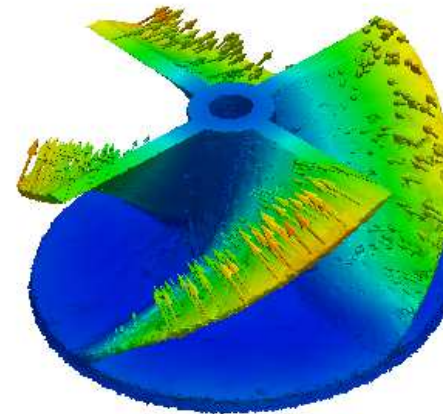
Macro-scale modeling: simulation results

24

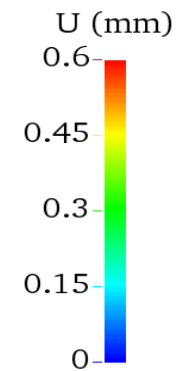
Final construction



$\|u\|$ in part and substrate



$\|u\|$ after removing the substrate



SPEIT collab.: bachelor thesis - Jian YANG

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► Macro-scale numerical modeling of SLM process by the super-layer approach

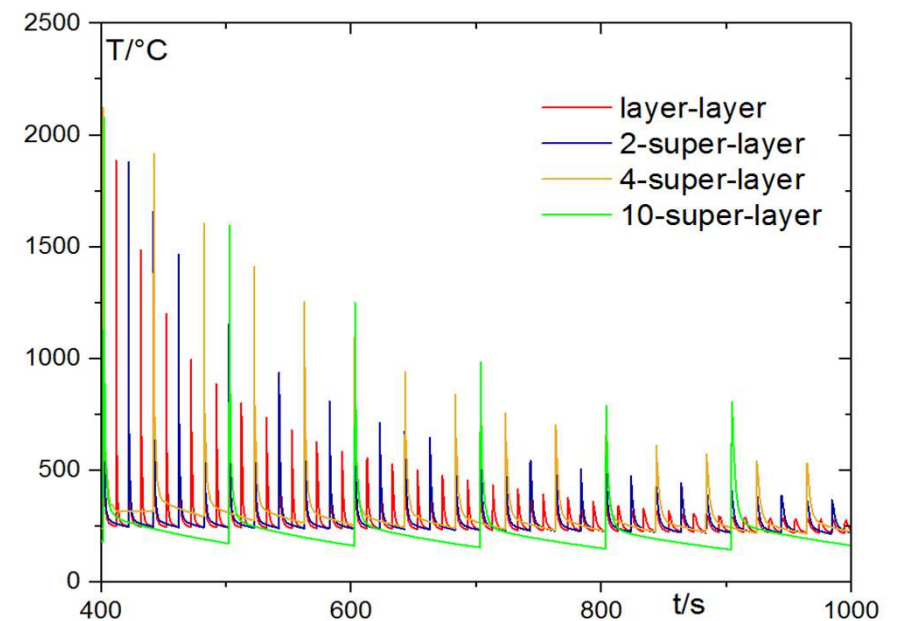
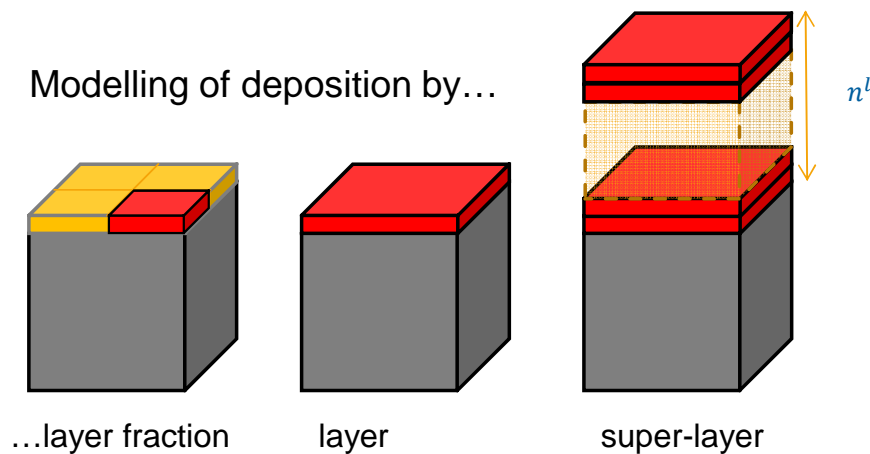
3-month internship at CEMEF, April - June 2018

Supervisors: Yancheng ZHANG and Michel BELLET (**CEMEF**), Shengyi ZHONG (**SPEIT**)

Experiments in SJTU, Modeling in CEMEF

- Reduce the calculation time of macroscale calculations
- Explore the validation domain of super-layer formulation

Second prize for the thesis poster.



SPEIT collab.: bachelor thesis – Constant Prassette

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► Inherent strain rate method for Direct Energy Deposition (DED)

3-month internship at CEMEF, April - June 2019

Supervisors: Yancheng ZHANG and Michel BELLET (**CEMEF**), Guanghua WEI (**SPEIT**)

- Implementation of Inherent strain rate in **CIMLIB®**
- Application to the process of DED

□ Implementation of inherent strain rate

Definition of inherent strain rate:

$$\dot{\epsilon} = \dot{\epsilon}^{el} + \dot{\epsilon}^{vp} + \dot{\epsilon}^{th}$$

$$\dot{\epsilon}^* = \dot{\epsilon}^{vp} = \dot{\epsilon} - \dot{\epsilon}^{el} - \dot{\epsilon}^{th}$$

Equations of equilibrium and mass conservation:

$$\begin{cases} \nabla \cdot \boldsymbol{\sigma} + \rho \mathbf{g} = 0 \\ \nabla \cdot \mathbf{u} = tr \dot{\epsilon}^{el} + tr \dot{\epsilon}^* + tr \dot{\epsilon}^{th} = -\frac{\dot{p}}{x} - 3\alpha(T)\dot{T} = 0 \end{cases}$$

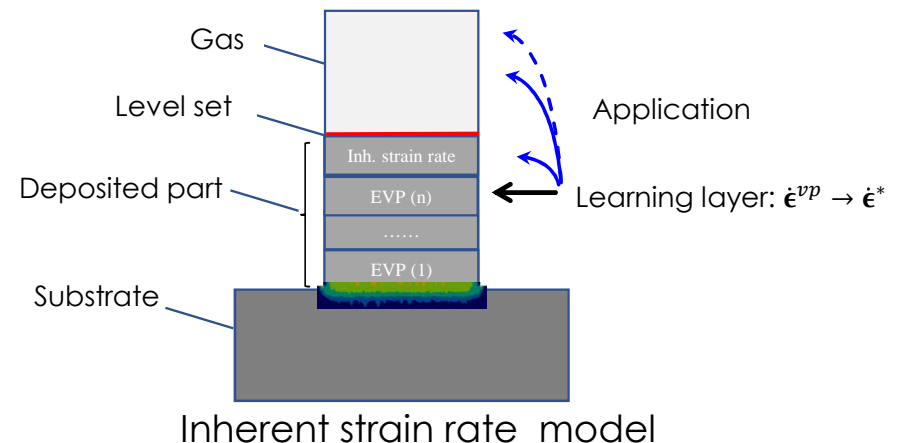
□ Adaptation to the process of DED

Problems:

- EVP calculation
- $\dot{\epsilon}^{vp}(\dot{\epsilon}^*)$ on substrate

Resolutions:

- Full calculations in first few layers
- Extraction ($\dot{\epsilon}^* = \dot{\epsilon}^{vp}$) and application after learning layer



SPEIT collab.: bachelor thesis – Constant Prassette

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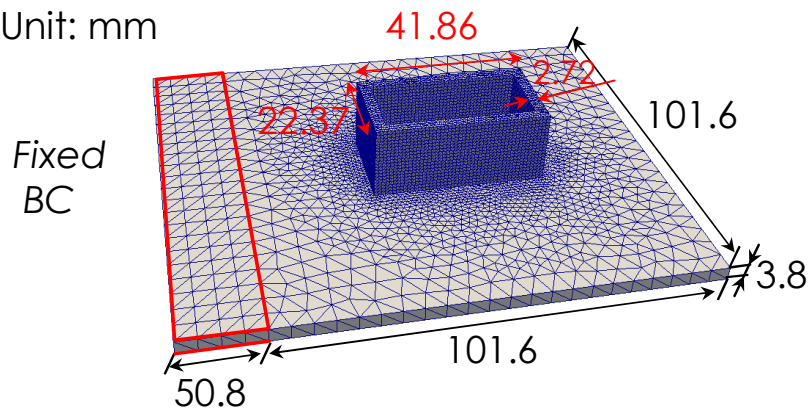
Application to the process of DED

Simulation system: 20 layers

-Number of element 108801

-Number of node 23829

Unit: mm



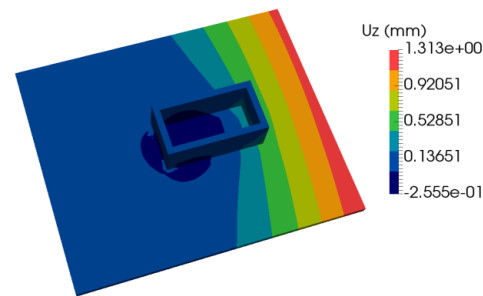
Simulation strategy:

10 EVP → Inh. strain rate → 10 Inh. strain rate

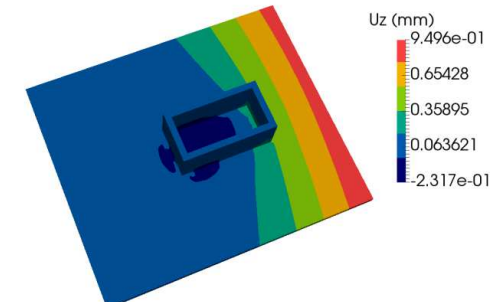
Simulation time:

-Complete calculation: 1h48

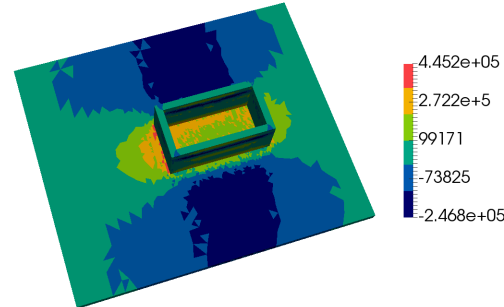
-Inherent strain : 1h25



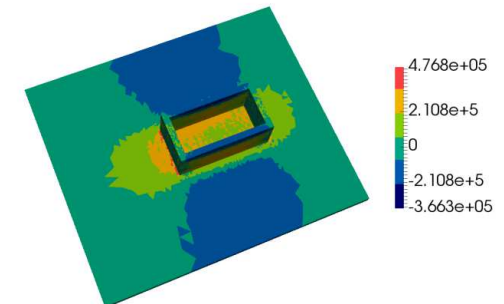
U_z (full EVP)



U_z (Inh. strain rate method)



σ_{xx} (full EVP)




σ_{xx} (Inh. Strain rate method)

Comparison of final distortion and residual stress for 20 layers deposition

Final conclusions:

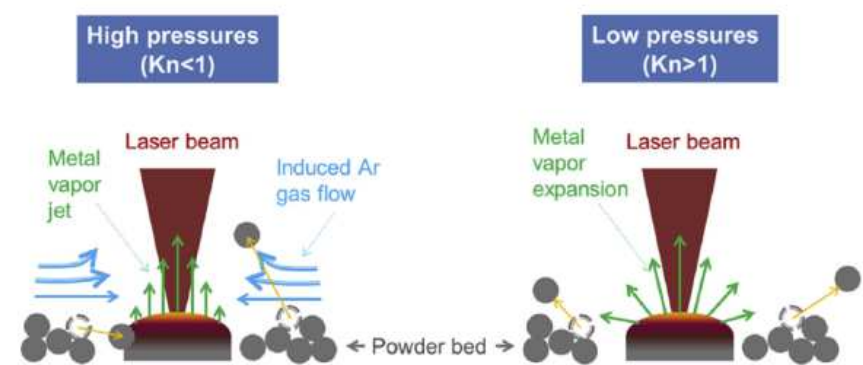
The thermo-mechanical analysis is performed for the **micro**-, **meso**- and **macro**-scale models under the **level-set** framework with **finite element** method:

	Micro-scale	Meso-scale	Macro-scale
Temperature	Melt pool, or track	Melt pool and Track	Whole part
Non-melted powder	Yes	Yes	Yes
Melt pool dynamics+shape	Yes		-
Mesh adaptation	Error estimation	Error estimation – total element number control	User defined – size control
Stress	Hot stress	Hot stress	Hot stress+ Residual stress
Distorsion	-	-	Yes
Complex geometry	-	-	Yes

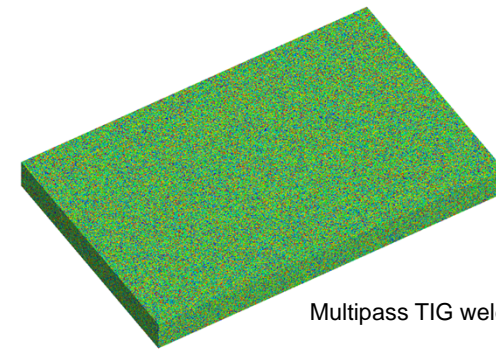
Small scale approaches: ongoing and future work

29

- ▶ **Hot tearing prediction**
- ▶ **Vaporization phenomena**
 - ▶ Impact on the shape of the fusion zone : recoil pressure, keyhole effect
 - ▶ Denudation process
- ▶ **Multitrack deposition**
 - ▶ Application to lattice structure details
- ▶ **Microstructure formation**
 - ▶ Modelling grain growth using CA-FE formulation (C.-A. Gandin and G. Guillemot at CEMEF)
- ▶ **Particle scale modelling**



Matthews et al., *Acta Materialia* (2016)



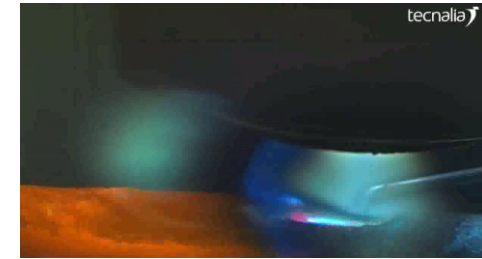
Multipass TIG welding (GTAW)

Chen, Guillemot, Gandin, Three-dimensional cellular automaton-finite element modeling of solidification grain structures for arc-welding processes, *Acta Materialia* (2016)

Macro-scale: ongoing and future work

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- ▶ **Interactions between the different scales**
- ▶ **Meso or particle scale → macroscale**
 - ▶ Inherent strain method, to make faster macroscale calculations
 - ▶ Superlayer formulation: design, validity
 - ▶ Model reduction
- ▶ **Macroscale → meso or particle scales**
 - ▶ Initialization of small scale calculations



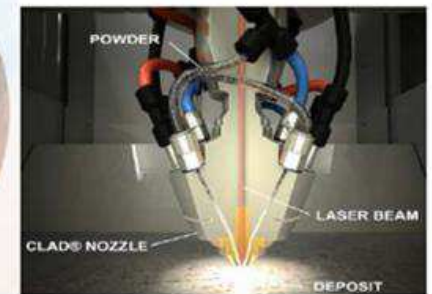
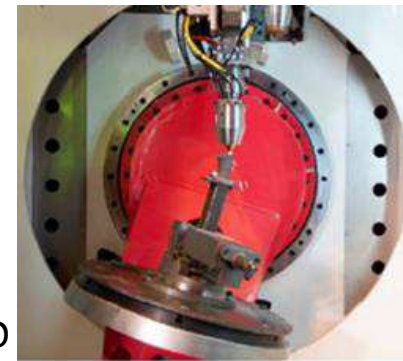
WAAM



... and also, all scales:

- ▶ **Extension to other additive manufacturing processes**
 - ▶ Wire arc metal additive manufacturing (WAAM)
 - ▶ Direct metal deposition (DMD)
 - ▶ Electron beam melting
 - ▶ ...

DMD



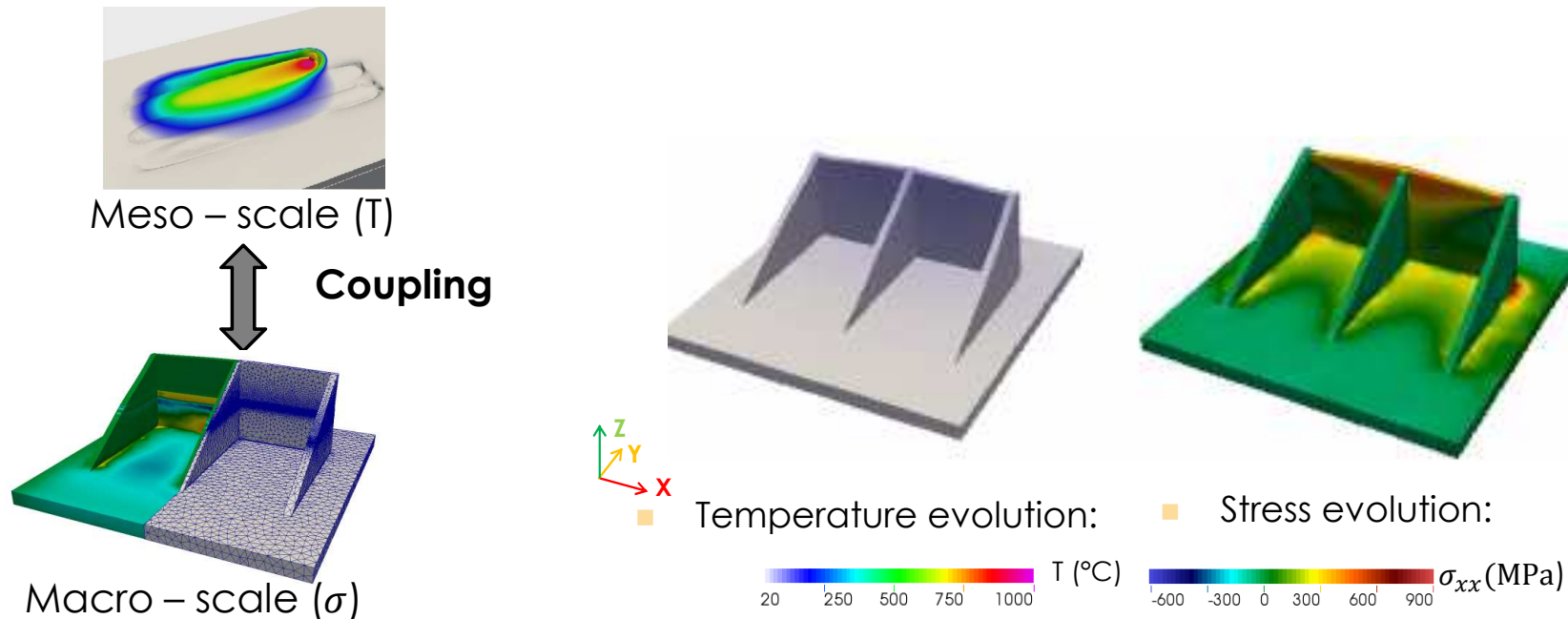
ongoing: Wire Arc Additive Manufacturing

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PhD Thesis MACCADAM: L. Ravix (2018 – 2021)

(co-supervised with: G. Guillemot, M. Bellet, Ch.-A. Gandin)

- **Controlled characteristics materials produced by WAAM (316L)**
 - Optimization of process parameters and building strategy
 - Coupling between and meso- and **macro-scale** approaches



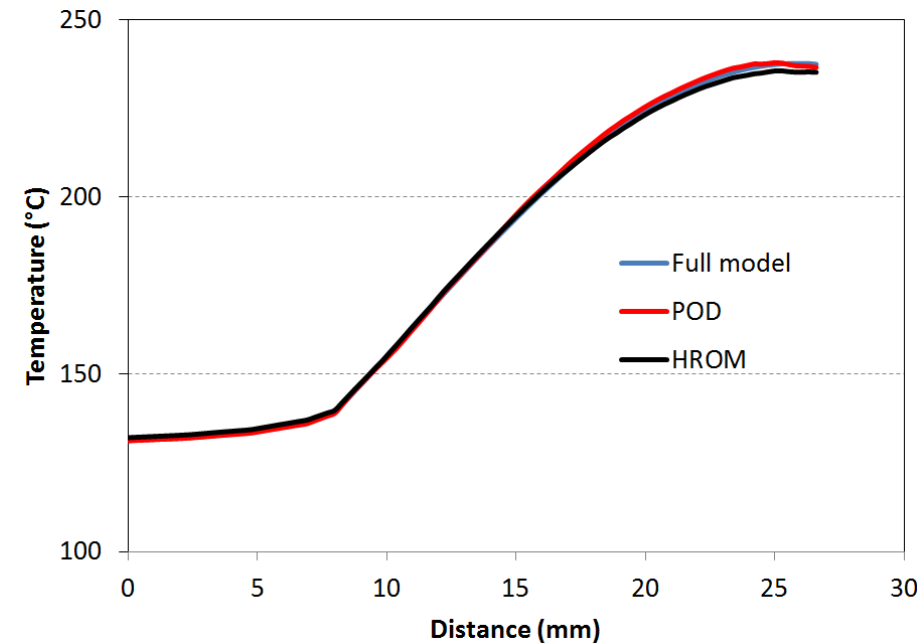
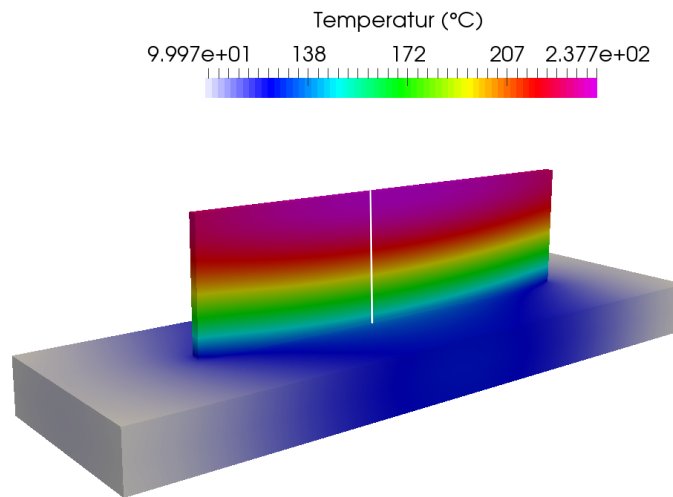
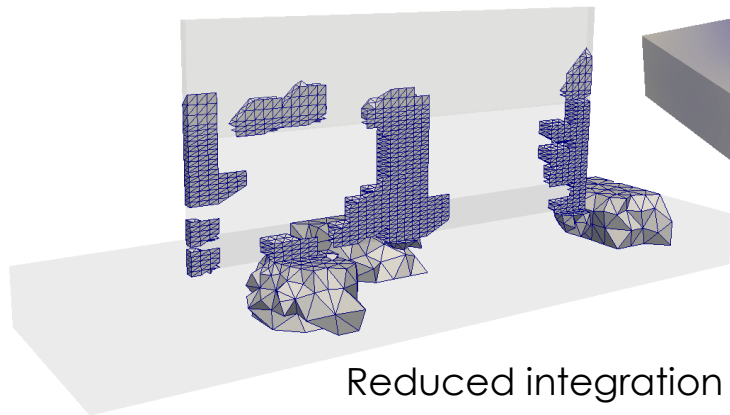
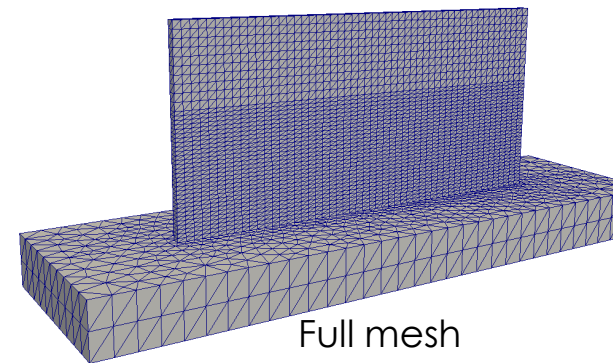
[LMGC]

ongoing: model reduction of direct energy deposition

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Thesis «Contrats Doctoraux Ecole»: J. Keumo Timatio (2018-2021)

(co-supervisors: M. Bellet and D. Rycklynck)



Hyper reduced-order model (316L)

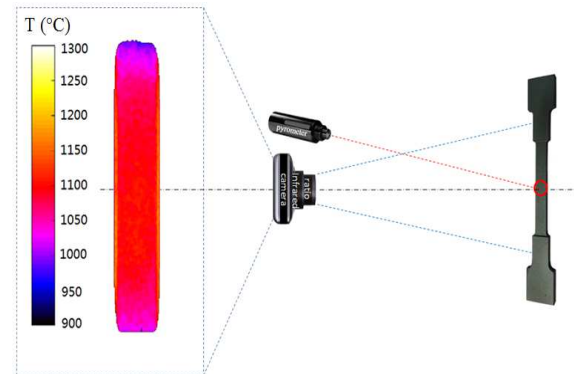
Ongoing: characterization of mechanical behavior in FA condition

33

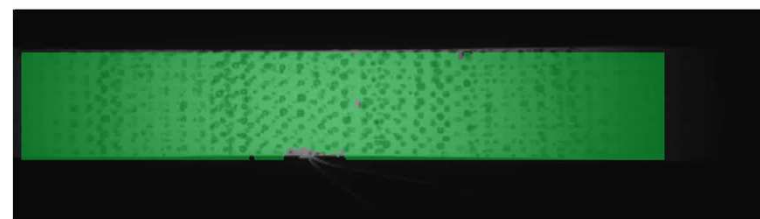
PhD Thesis SAFRAN and CSC: F. Gao (2017-2021)

(co-supervisors: M. Bellet)

- ▶ **Determine the material behavior at very high temperature (near fusion)**
- ▶ **Machine « Dedimet » developed at CEMEF**
 - Traction /relaxation tests with resistive heating (**Joule effect**), under **vacuum or controlled atmosphere**.
 - Infrared imaging **temperature field** (+ thermocouples, + pyrometers)
 - ▶ **Displacement field** by correlation image (laser speckle interferometry)
 - **Coupled electrical/thermal/mechanical modelling**, implemented in CIMLIB®
 - **Identification by inverse method**
 - **Characterization of anisotropic elasto-viscoplasticity**
 - **Ti-6Al-4V** and **In718** alloys



Non-contact temperature measurement



Displacement measurement at 1000 °C



Dedimet machine



Questions ?

